

Predicting and Avoiding Die Attach, Wire Bond, and Solder Joint Failures

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Why Die Attach/Wire Bonds/Solder Joints?

- Interconnects tend to be the primary driver for reliability of power modules
 - Influenced by the robustness
 of today's power devices
 (IGBT and FET, Si/GaN/SiC) and the tendency
 to towards conservative design (derating margin)

Active endurance tests of direct-bonded-copper based MOSFET power modules showed that <u>the aluminum bonds</u> and <u>the die attach</u> are the most critical design elements regarding product reliability – Robert Bosch, 2016

Bondwire

Base Plate

Bond

Substrate

Bond

Diode

Chip Solder

Base Plate Solder

Thermal Grease

Bond IGBT

How to Avoid Die Attach, Wire Bond Failures?

• Step 1: Good quality control

 Step 2: Perform reliability prediction using physics of failure (PoF)



What is Physics of Failure (PoF)?

• Also known as reliability physics

• <u>Common Definition</u>:

 The process of using modeling and simulation based on the fundamentals of physical science (physics, chemistry, material science, mechanics, etc.) to predict reliability and prevent failures







Physics of Failure: Modeling and Simulation

- What are we modeling / simulating?
- Reliability (t > 0) = Material Change or Material Movement

Fundamental Material Mechanisms

- Diffusion
- Oxidation/Reduction
- Creep
- Fatigue

Material Movement: Diffusion

- Motion of electrons, atoms, ions, or vacancies through a 0 material
 - Typically driven by a concentration gradient (Fick's Law)

$$\begin{aligned} \mathbf{J}_{\mathbf{A}}(\mathbf{x},t) &= -\mathbf{D}_{\mathbf{A}} \frac{\partial \mathbf{C}_{\mathbf{A}}(\mathbf{x},t)}{\partial \mathbf{x}} \\ n\left(x,t\right) &= n(0) \left[1 - 2\left(\frac{x}{2\sqrt{Dt\pi}}\right)\right] \end{aligned} \overset{\text{for all }}{\to} \mathbf{D}_{\mathbf{A}} \overset{\text{f$$

Can be driven by other forces (electromotive force, stress) 0

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(X)

Material Movement: Creep

- The tendency of a solid to permanently deform when subjected to a fixed load
 - Corollary: Tendency of a solid to relieve stress when loaded at a fixed displacement
- Metals: Driven by movement of defects within the crystalline structure
 - Dislocations (edge or screw)
 - Grain Boundaries



Material Movement and M&S

- All physics of failure models can be condensed into answers to three questions
 - How large is the stress?
 - At what rate is this stress driving material movement?
 - At what time will this material movement induce failure?

PoF-Based Reliability Prediction

- Most physics-of-failure (PoF) based models are semi-empirical
 - The basic concept is still valid
 - Requires calibration
- Calibration testing should be performed over several orders of magnitudes
 - Allows for the derivation of semi-empirical constants, if necessary
- The purpose of PoF is to limit, but not eliminate, the influence of material and geometric parameters
 - E.g., Solder: Testing must be re-performed for each package family (ball array devices, gullwing, leadless, etc.)

How to Perform PoF? Stress/Strain/Energy

 One of the key differentiations in PoF techniques is how to capture the stress/strain/energy within the interconnect



$$\boxed{\epsilon = 6 \left(\frac{r}{D}\right) \left(\frac{L}{D} - 1\right)^{0.5} \left[2\alpha_{base} + \frac{\alpha_{base} - \alpha_{wire}}{1 - (D/L)}\right] \Delta T}$$

Option 2: Analytical Equations

Option 1: Finite element model

- Most manufacturers will use Option 1 at some point during the development process (especially for complex geometries)
 - However, Option 2 is preferred for tradeoff analysis and for users of the power modules
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How to Perform PoF? Damage Accumulation

 Lots of discussion about key damage parameter (stress, strain, elastic, plastic, plasticity, creep, energy)

$$\frac{\Delta\varepsilon}{2} = \frac{\sigma'_f}{E} (2N_f)^b + \varepsilon'_f (2N_f)^c$$

Basquin (high-cycle) Coffin-Manson (low-cycle)

$$\frac{da}{dN} = C\Delta K^{m} \qquad \qquad N_{0} = C_{1} \times (\Delta W_{a}ve)^{C_{2}}$$
Paris
(high-cycle)
$$\frac{da}{dN} = C_{3} \times (\Delta W_{ave})^{C_{4}}$$
Darveaux
(low-cycle) DfR Solutions

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How to Perform PoF? What's Important?

- One of the most critical aspects of PoF is including primary parameters and excluding secondary effects
 - Can sometimes drive the semi-empirical nature of PoF







Predicting Reliability of Die Attach



Predicting the Reliability of Die Attach

- Die attach tends to have only one failure mode
 - Thermal cycle fatigue due to power cycling
- Frequency of the power cycle can play a very critical role
 - If the power cycle frequency is high enough, the failure site will shift from the die attach to the wire bond (thermal inertia)
- Key Challenge: Defining failure
 - Most die attach configurations do not conduct electricity
 - Die attach is primarily a thermal path

Die Attach Fatigue (Englemaier, 1982)

$$\Delta \gamma = \frac{(\sqrt{L_d^2 + W_d^2})(CTE_{die} - CTE_{DBC})\Delta T}{2h}$$

Strain range at the die/die attach interface

- h = die thickness
- W = die width
- L = die length

- α = coefficient of thermal expansion (CTE)
- ∆T = change in temperature

Die Attach Fatigue (Time to Failure)

$$N_f = \varepsilon_f (\Delta \gamma)^c$$

- Coffin-Mason based low-cycle fatigue damage model
- Tin-based solders (SnPb, SAC305, Sn3.5Ag, etc.) tend to have fatigue exponents around 2 to 2.5
 - SnPb: ~2; Sn3.5Ag: 2.2; SAC305: ~2.4
- Fatigue exponents for new die attach solders have not yet been widely validated
 - Nanosilver, BiAgX, Sn25Ag10Sb ("J" alloy)

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Nanosilver: Is it Better Than Solder?

- Sintering with nanosilver is challenging (and expensive)
 - Requires pressure
 - For large die, post sintering stress relief step is often required

• Voiding must be controlled



Fig. 8 Plot of characteristic lifetime of sintered joints during thermal cycling vs. density.



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Nanosilver: Is it Better Than Solder? (cont.)

• For the most part, the industry agrees nanosilver is more reliable, but there can be issues



M. Beierlein, Adv. Pack. Conf. 2013 DfR Solutions

L. Melchor, Doctoral Dissertation

Nanosilver: Is it Better Than Solder? (cont.)

 Different publications provide different values for fatigue constants and exponents

Author	Constant	Value	Exponent
M. Knoerr	1.6E-01	Plastic Strain	-3.0
Y. Tan	5.8E+11	Shear Stress	-9.4
X. Li	1.6E-09	Shear Strain	-7.6

$$N_{f,\text{Sn96},\text{SAg3Cu0},\text{S}} = 0,075 \cdot \Delta \varepsilon_{\text{plastic}}^{-1,79}$$

$$N_{f,\text{Nano-Silver}} = 0,160 \cdot \Delta \varepsilon_{\text{plastic}}^{-2,90}$$



Y. Tan, et. al., Conf. on Fracture, 2013

Vibration Fatigue

- Lifetime under mechanical cycling is divided into two regimes
 - □ Low cycle fatigue (LCF)
 - □ High cycle fatigue (HCF)
- LCF is driven by inelastic strain (Coffin-Manson) $\varepsilon_p = \varepsilon_f (2N_f)^c$

-0.5 < c < -0.7; 1.4 < -¹/c > 2

 HCF is driven by elastic strain (Basquin)

$$\mathcal{E}_{e} = \frac{O_{f}}{E} (2N_{f})^{b}$$

-0.05 < b < -0.12; 8 > -¹/b > 20



Fig.6.19: Total strain range as the sum of the plastic and the elastic strain range. Material: AISI 4340 (annealed) [19]. (Range $\Delta \epsilon \approx 2 \cdot \epsilon_n$).





Predicting Reliability of Wire Bonds



Reliability: When do Wire Bonds Fail?

- Exposure to elevated temperature
 - Intermetallic formation
- Exposure to elevated temperature/humidity
 - Corrosion
- Exposure to temperature cycling
 - Low cycle fatigue



Reliability at Elevated Temperatures

- Not an issue in aluminum-aluminum wire bond system
 - The lack of intermetallic formation and differential diffusion makes it relatively immune to purple plague
 - Prior studies have found little change in resistance after 1000 hours at 300C
- Bigger issue in mixed metal systems, like gold-aluminum
 - Formation of brittle AuAl2 (purple plague) at 350C
 - Diffusion of gold into Au₅Al₂ causes
 Kirkendall voiding at lower temps





Reliability Prediction (Elevated Temperature)

- Is an absolute reliability prediction of wire bond reliability at elevated temperature possible?
- Short answer: NO
 - Diffusion behavior is very sensitive to bonding temperature, quality of bond, aluminum alloy, aluminum bond pad thickness, and encapsulant chemistry
 - Low bonding temperature
 - Si in Al-Cu bond pad
 - Thin bond pad (~1 um)
 - Bromide-free flame retardants

 Can change absolute and relative (acceleration factor) time to failure

Reliability Prediction – Temperature (cont.)

 For gold-aluminum, prediction is primarily by extrapolation from test results using Arrhenius and a conservative activation energy (0.9 eV)

$$t_f = A \exp\left(\frac{\Delta H}{kT}\right)$$

- However, there is some question as to the presence of a minimum temperature
 - Periodically reported as 125C for unencapsulated and 85C for encapsulated
 - Observed in other systems (tin-copper and whiskers)

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Other Systems (Cu-Al)

- Copper-aluminum forms intermetallics at a much slower rate
 HJ Kim, IEEE CPT, 2003
 - $_{\circ}$ Most common activation energy of 1.26 1.47 eV
 - Micron reported 0.63 eV L. England, ECTC, 2007
- Molding compound has little effect



Intermetallic Penetration - AlSi-Cu Bond (aged 800 hours at 180°C)



Intermetallic Penetration - AlSi-Au Bond (aged 200 hours at 200°C)



C. Breach, The Great Debate: Copper vs. Gold Ball Bonding



L Levine, Update on High Volume Copper Ball Bonding

Other Systems (Cu-Al)(cont.)



Figure 6. Ball Pull Strength after HTS at 200°C

- Cu-Al can show improved performance over Au-Al
 - Not to the extent expected based on intermetallic growth
 - Different failure mode (gradual vs. sudden)

Shear Strength at Elev Temp



Shear strength of Au and Cu ball bonds on Al pads
At lower temperatures (<150C) they are similar in strength loss

J. Onuki, M. Koizumi, I. Araki. IEEE Trans. On Comp. Hybrids & Manfg. Tech. 12 (1987) 550

Cu-Al and Elevated Temperature – Concerns

- Different intermetallics form at different temperatures
 - Can a 150C/200C test be extrapolated to 85C?
 - Fracture mode with pure Cu changed from bulk Cu to interfacial failure
- Some indications that oxidation of the wedge bond may be a critical weak point









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S. Na, T. Hwang, J. Kim, H. Yoo, and C. Lee, "Characterization of IMC growth in Cu wire ball bonding on Al pad

metallization", ECTC, IEEE, 2011

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Wire Bonds and Temperature/Humidity

- Degradation primarily occurs in the gold wire bond / aluminum bond pad system
- Driven by galvanic corrosion
 - Absence of galvanic couple in aluminum/aluminum and copper/aluminum systems tends to limit corrosive behavior
- Presence of halides, especially chloride, can accelerate corrosive behavior
 - Getters in the molding compound tend to reduce this risk

Wire Bonds and Temperature/Humidity (cont.)

- The mechanism of Au-Al corrosion has been primarily described by Peck's Law
- Peck's Law for Temperature and Humidity
 - Empirical study of THB/HAST (85/85, 110/85 & 5-70 VDC)

$$t_{life} = A_0 R H^{-n} f(v) \exp\left(-\frac{E_a}{kT}\right)$$

 t_{life} = time to failure, RH = relative humidity, E_{a} = activation energy (0.79eV), T = temperature, $A_0 = material constant$

- n = empirical constant (2.66)
- k = Boltzmann constant
- f(v) = voltage function (power law, ~1.5)

Copper Wire Bond and Temperature/Humidity

• Copper is not as noble as gold

- Noble coatings (palladium) can come off during bonding
- Palladium (Pd) coating can also create galvanic couple with copper
- Studies have shown early failures during temp/humidity testing
 - Some dependency on molding compound (need lower pH, lower halogen content)
 - Uncertain if JEDEC test with acceleration factor based on Peck's equation (based on aluminum/gold) is still valid



Figure 6. Failure rates after 336 hours biased HAST test for three experimental mold compound formulations with 0.8 mil bare Cu and Pd-coated Cu wire on Al bond pads. (5V bias, 130oC, 85% RH, electrical open/short

H. Clauberg, Chip Scale Review, Dec 2010



Copper Wire Bond and Temperature/Humidity (cont.)

- T. Boettcher believe early failures are due to galvanic corrosion of Cu-rich intermetallics (Cu₉Al₄) (EPTC 2010)
 - Induces the formation of copper oxides between the intermetallic and the copper bond wire
- Initial failures during JEDEC HTRB and Autoclave testing were reversed by increasing the amount of intermetallic through annealing
 - Small anode (intermetallic) relative to cathode greatly increases corrosion rate



Status of Cu-WB Durability-Reliability Research iNEMI P2B

- iNEMI Phase 2B: HAST 130degC/85% Rel. Humidity for 384 hours on loose parts
 - Tested every 4 days at 96, 192, 288 and 384 hours.
 - Failure pattern suggests more of a durability issue (however, how relevant is 384 hours under HAST?)

	96hrs	192hrs	288hrs	384hrs
BGA1	0/24	0/24	0/24	0/24
BGA2	0/34	0/34	1/34	1/33
BGA3	1/34	0/33	0/33	1/33
BGA4	0/34	0/34	0/34	2/34
BGA5	0/34	0/34	0/34	1/34
BGA7	0/96	0/96	0/96	0/96

Fig14. HAST 130degC/85%RH Test Result for BGA

Wire Bonds and Temperature Cycling (Wedge Flexure)

- Driven by differences in coefficient of thermal expansion (CTE)
 - Flexing motion results create microcracks at the heel of the wirebond
 - Model based on theory of curved beams

$$\varepsilon = 6 \left(\frac{r}{D}\right) \left(\frac{L}{D} - 1\right)^{0.5} \left[2\alpha_{base} + \frac{\alpha_{base} - \alpha_{wire}}{1 - (D/L)}\right] \Delta T$$

Strain at the heel of the wire (assumes bond pads at same height)

- r = wire radius
- D = half wire span
- L = wire length

- α = coefficient of thermal expansion (CTE)
- ΔT = change in temperature

Wedge Flexure (cont.)

Coffin-Mason Based Low-Cycle Fatigue

$$N_f = C \varepsilon^m$$

• **C** and **m** empirically determined to be 1.0 and (-1.4) respectively for aluminum wedge bonds



Wire Bonds and Temperature Cycling (Axial Tension)

 Wire encapsulated in molding compound can experience tensile stresses in the wire due to differential expansion and contraction

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$$\sigma_w = E_w (\alpha_e - \alpha_w) \Delta T$$

Modulus of molding compound is ignored because of its minimal contribution

$$N_f = C\sigma^m$$



Wire Bonds and Temperature Cycling (Shear)

 Shear stresses between the substrate (s), the bond pad (p) and the wire (w)

$$\tau_{w,M} = \left\{ \frac{r^2}{4Z^2 A_w^2} \left[\frac{\cosh(Zx_w)}{\cosh(Zl_w)} - 1 \right]^2 + \frac{\sinh^2(Zx_w)}{\cosh^2(Zl_w)Q^2} \right\}^{1/2} \Delta T$$

- r is wire radius
- A is cross-sectional area
- W is width
- G is shear modulus
- o **b** is thickness
- I is length

$$Z^{2} = \frac{G_{p}}{b_{p}} \left[\frac{r}{E_{w}A_{w}} + \frac{(1-\nu_{s})W_{p}}{E_{s}A_{s}} \right]$$

$$Q = \frac{G_p}{b_p Z} \left[(\alpha_w - \alpha_s) + \frac{(\alpha_s - \alpha_p)}{1 + (E_s A_s) / [E_p A_p (1 - \nu_s)]} \right]$$

r----

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Competing Failure Mechanisms



Validation (Aluminum-Aluminum System)



Copper Wire and Temperature Cycling

- Power module industry believes copper wire is more robust than aluminum
 - Changes being implemented for electric drivetrain
- Part of improvement is believed to be due to reduced temperature variation from improved thermal conductivity
- Part of improvement could be due to recrystallization N. Tanabe, Journal de Physique IV, 1995
 - Can result in self-healing



Figure 5: Temperature distribution along a 400µm wire of 12mm length, Al dashed curve, Cu solid red curve, current is 19A

D. Siepe, CIPS 2010

• Part of improvement could be more robust fatigue behavior



Copper vs. Gold – Temperature Cycling

Copper superior based on these publications







Fig. 9. Experimental results of fatigue tests in the S-N diagram for three values of mean stress. Specimens marked with an arrow did not fail.

G. Pasquale, J. Microelectromech Sys.,, 2011



Aluminum vs. Copper – Temperature Cycling





Thermal Cycling Reliability and Assembled Parts

- Loose Cu wire-bonded parts have passed component level thermal cycle tests, but have failed during thermal cycling of automotive E/E modules
- Believed, but not confirmed, to be due to additional expansion-contraction stresses from the CTE mismatch part and PCB







Predicting Reliability of Solder Joints



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Solder Joint Fatigue

 Knowing the critical drivers for solder joint fatigue, we can develop predictive models and design rules





Predictive Models – Physics of Failure (PoF)

- Modified Engelmaier for Pb-free Solder (SAC305)
 - Semi-empirical analytical approach
 - Energy based fatigue
- Determine the strain range ($\Delta\gamma$)

$$\Delta \gamma = C \frac{L_D}{h_s} \Delta \alpha \Delta T$$

• C is a correction factor that is a function of dwell time and temperature, L_D is <u>diagonal distance</u>, α is coefficient of thermal expansion (<u>CTE</u>), Δ T is temperature cycle, h is <u>solder joint height</u>

Predictive Models – Physics of Failure (PoF)(cont.)

• Determine the shear force applied to the solder joint

$$\left(\alpha_2 - \alpha_1\right) \cdot \Delta T \cdot L = F \cdot \left(\frac{L}{E_1 A_1} + \frac{L}{E_2 A_2} + \frac{h_s}{A_s G_s} + \frac{h_c}{A_c G_c} + \left(\frac{2 - \nu}{9 \cdot G_b a}\right)\right)$$

- F is shear force, L is <u>length</u>, E is <u>elastic modulus</u>, A is the area, h is thickness, G is shear modulus, and a is edge length of bond pad
- Subscripts: 1 is <u>component</u>, 2 is <u>board</u>, s is solder joint, c is bond pad, and b is board
- Takes into consideration foundation stiffness and both shear and axial loads



Predictive Models – Physics of Failure (PoF)(cont.)

Determine the strain energy dissipated by the solder joint

$$\Delta W = 0.5 \cdot \Delta \gamma \cdot \frac{F}{A_s}$$

 Calculate cycles-to-failure (N₅₀), using energy based fatigue models

$$N_f = (0.0019 \cdot \Delta W)^{-1}$$

Solder Joint Validation

- Energy-based analytical equation shows strong correlation to both test and field failures
- When correlation is not observed, typically driven by the presence of an axial loading condition (constraints, potting)
 - Requires use of compatibility of displacements





Summary / Conclusion

- The field of reliability prediction is not stagnant
 - Driven by need for new materials, new technologies
 - Driven by demand for faster time to market (can not test everything)
 - Driven by limited resources (can not FEA everything!)
- Be aware when knowledge is sufficient, validated by a physical understanding and testing, to proceed with modeling and simulation